






Article

Ab-Initio Study of Magnetically Intercalated Platinum Diselenide: The Impact of Platinum Vacancies

Peter D. Reyntjens ^{1,2,3} , Sabyasachi Tiwari ^{1,2,3} , Maarten L. Van de Put ¹ , Bart Sorée ^{3,4,5}  and William G. Vandenberghe ^{1,*} 

- ¹ Department of Materials Science and Engineering, The University of Texas at Dallas, 800 W Campbell Rd, Richardson, TX 75080, USA; peter.reyntjens@utdallas.edu (P.D.R.); sabyasachi.tiwari@utdallas.edu (S.T.); maarten.vandepu@utdallas.edu (M.L.V.d.P.)
- ² Department of Materials Engineering, KU Leuven, Kasteelpark Arenberg 44/Box 2450, 3001 Leuven, Belgium
- ³ IMEC, Kapeldreef 75, 3001 Leuven, Belgium; bart.soree@imec.be
- ⁴ Department of Electrical Engineering, KU Leuven, Kasteelpark 10, 3001 Leuven, Belgium
- ⁵ Department of Physics, University of Antwerp, Groenenborgerlaan 161, 2020 Antwerp, Belgium
- * Correspondence: william.vandenberghe@utdallas.edu; Tel.: +1-972-883-5793

Abstract: We study the magnetic properties of platinum diselenide (PtSe₂) intercalated with Ti, V, Cr, and Mn, using first-principle density functional theory (DFT) calculations and Monte Carlo (MC) simulations. First, we present the equilibrium position of intercalants in PtSe₂ obtained from the DFT calculations. Next, we present the magnetic groundstates for each of the intercalants in PtSe₂ along with their critical temperature. We show that Ti intercalants result in an in-plane AFM and out-of-plane FM groundstate, whereas Mn intercalant results in in-plane FM and out-of-plane AFM. V intercalants result in an FM groundstate both in the in-plane and the out-of-plane direction, whereas Cr results in an AFM groundstate both in the in-plane and the out-of-plane direction. We find a critical temperature of <0.01 K, 111 K, 133 K, and 68 K for Ti, V, Cr, and Mn intercalants at a 7.5% intercalation, respectively. In the presence of Pt vacancies, we obtain critical temperatures of 63 K, 32 K, 221 K, and 45 K for Ti, V, Cr, and Mn-intercalated PtSe₂, respectively. We show that Pt vacancies can change the magnetic groundstate as well as the critical temperature of intercalated PtSe₂, suggesting that the magnetic groundstate in intercalated PtSe₂ can be controlled via defect engineering.

Keywords: transition metal dichalcogenides; magnetism; Monte Carlo



Citation: Reyntjens, P.D.; Tiwari, S.; Van de Put, M.L.; Sorée, B.; Vandenberghe, W.G. Ab-Initio Study of Magnetically Intercalated Platinum Diselenide: The Impact of Platinum Vacancies. *Materials* **2021**, *14*, 4167. <https://doi.org/10.3390/ma14154167>

Academic Editor: Mirko Poljak

Received: 29 May 2021

Accepted: 20 July 2021

Published: 27 July 2021

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Copyright: © 2021 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

1. Introduction

The field of two-dimensional (2D) spintronics [1] has seen unprecedented attention in the past few years, thanks to the recent experimental discovery of two-dimensional (2D) magnetic materials CrI₃ [2,3] and CrGeTe₃ [4]. However, the low Curie temperature of experimentally discovered 2D magnets (45 K for monolayer and 61 K for bulk CrI₃) impedes their technological application. Thankfully, there are many possible 2D magnets. One avenue of finding such 2D magnets is searching the space of magnetic crystals [5,6]. Another avenue of realizing 2D magnets is through magnetic doping [7,8] of conventional 2D materials. The advantage of such magnetically doped magnets is the ability to control their properties through charge transfer.

Transition metal dichalcogenides (TMDs) offer a promising avenue for realizing 2D magnets through magnetic doping [7–13]. There have been many theoretical [7,8,11,12,14] and experimental [9,10,13] reports of realizing 2D magnetism in semiconducting TMDs through metal doping. On the theoretical side, Mishra et al. [11] investigated the effect of Mn doping on the magnetic properties of MoS₂, MoSe₂, MoTe₂, and WS₂. Mn-doped MoS₂ was also studied by A. Ramasubramaniam and D. Naveh [12], who obtained promising results predicting magnetic order persisting above room temperature. The work of Muhammad H. et al. [9] combined experimental fabrication and characterization with DFT

methods to investigate the magnetic properties of Ni-doped WSe₂, which they reported to have room-temperature magnetic ordering. Luo et al. produced a review paper discussing the doping and functionalization of several different 2D TMDs [15], including MoS₂, MoSe₂, WS₂, and WSe₂. However, similar work on realizing magnetic order in metallic TMDs with the 1T structure is not as common. As recent reports on 2D metallic magnetic systems, e.g., Fe₃GeTe₂ [16], show room-temperature magnetic order, it is natural to look at other metallic systems, such as PtSe₂.

Metallic TMDs are interesting because free electrons can provide a pathway for long-range magnetic interaction. One such interesting metallic TMD is PtSe₂. PtSe₂ forms a layered TMD with semimetallic character [17]. More recently, epitaxial growth of mono- and few-layer PtSe₂ has revealed a transition to semiconducting character for the monolayer and bilayer PtSe₂ [18,19]. Moreover, the large atomic weight of PtSe₂ can result in a higher anisotropy, which is necessary for the existence of 2D magnetic order [20]. Kar et al. performed a first-principles study of the magnetism in doped PtSe₂ monolayers and found promising higher-than-room-temperature magnetic ordering [21]. However, detailed work on the magnetic ordering, including Monte Carlo simulations for the estimation of the transition temperature in magnetically doped PtSe₂, is still missing from the literature. The presence of Pt vacancies in pristine PtSe₂ is responsible for a spin polarization of the electronic cloud around the vacancy, making defective PtSe₂ a material of interest for future spintronic applications [22,23]. Moreover, theoretical studies have shown that the presence of vacancies in TMDs is energetically favorable in pristine PtSe₂ grown in Se-rich conditions [24] and reduces the formation energy of intercalated systems [8]. In Ref. [24], the authors find a Pt vacancy density of $4.3 \pm 1.4 \times 10^{12} / \text{cm}^2$ in ultrathin layered PtSe₂.

In this work, we theoretically investigate the magnetic order in bulk PtSe₂ intercalated with Ti, V, Cr, and Mn, using first-principles density functional theory (DFT) calculations and Monte Carlo (MC) simulations. In Section 2, we describe the methods used in our work, starting from DFT for the calculation of structural parameters and magnetic groundstates, up to the critical temperature calculation using Monte Carlo. In Section 3, we first present the equilibrium position of intercalants in PtSe₂ obtained from the DFT calculations. Next, we present the magnetic groundstates for each of the intercalants in PtSe₂ along with their critical temperature. We study the effect of Pt vacancies on the structure and formation energy and investigate their impact on the magnetic order. Finally, in Section 4, we conclude.

2. Methods

Figure 1a illustrates our computational model. We first intercalate a $2 \times 2 \times 2$ supercell of PtSe₂ and calculate the total energy of various magnetic configurations including ferromagnetic (FM) and anti-ferromagnetic (AFM) configurations using DFT+*U* calculations. Details on the computational parameters are included in Appendix A. To take into account the magnetic anisotropy, we perform total energy calculations with spin-axis oriented in the in-plane and the out-of-plane direction.

Next, we model the magnetic structure of an intercalated PtSe₂ supercell using a parameterized Heisenberg Hamiltonian:

$$H = \frac{1}{2} \sum_{i,j} S_i J_{ij} S_j + \sum_i D (S_i^z)^2, \quad (1)$$

where $S = S^x \hat{x} + S^y \hat{y} + S^z \hat{z}$ is the magnetic moment vector of the intercalant atom. The exchange interaction strength tensor [7], with elements J_{ij} describing the strength of the interaction between spins at site i and j , is assumed to be rotationally invariant in the direction of the PtSe₂ planes, because PtSe₂ has in-plane rotational symmetry. Additionally, we assume that the interaction strength in the x and the y directions is equal and that the tensor's diagonal elements are vanishingly small. The result is a diagonal exchange interaction strength tensor with elements J_{ij} where $J_{ij}^x = J_{ij}^y$. We consider the range of interaction

up to the nearest-neighbor atoms in the in-plane and the out-of-plane direction shown in Figure 1b. The second term is the single-ion anisotropy D . With J_{ij} being diagonal and rotationally invariant in the plane of the PtSe₂ layers, the Heisenberg Hamiltonian becomes

$$H = \frac{1}{2} \sum_{i,j} \left(S_i^z J_{ij}^z S_j^z + S_i^x J_{ij}^x S_j^x \right) + \sum_i D (S_i^z)^2, \quad (2)$$

Here, x/z is the direction of spins, when oriented in the in-plane/out-of-plane direction, as shown in Figure 1c. The parameters $J_{ij}^{x/z}$ and D are obtained by fitting to the total-energy DFT calculations using the method developed in Ref. [25]. To assess the validity of the nearest-neighbor approximation, we have performed additional calculations to investigate the effect of the second-nearest-neighbor interactions, of which the details are explained in Section S1 of the Supplementary Information.

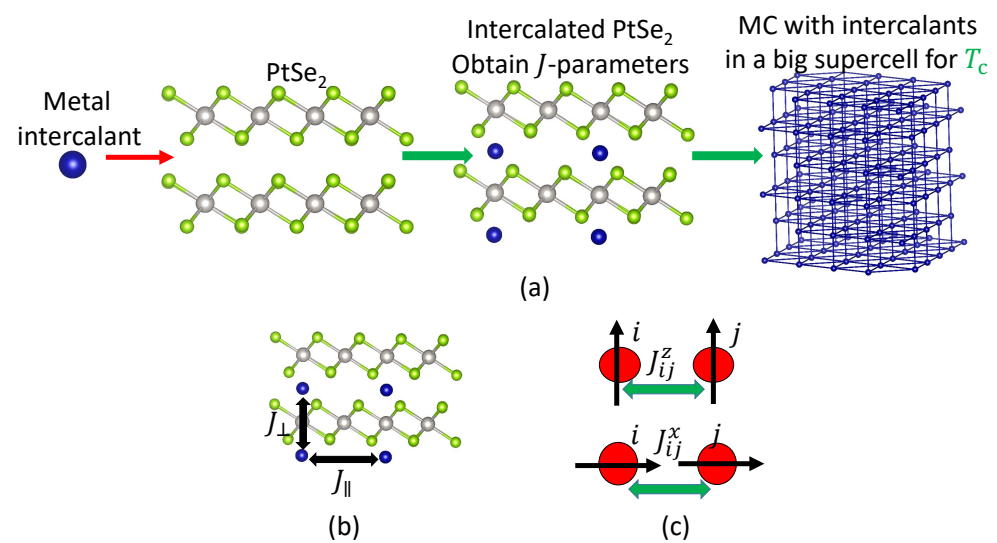


Figure 1. (a) The magnetic ions are intercalated in PtSe₂. (b) The exchange parameters in the in-plane (J_{\parallel}) and in the out-of-plane (J_{\perp}) direction. (c) The J parameters between spins i and j oriented in the in-plane direction (J_{ij}^x) and in the out-of-plane direction (J_{ij}^z).

After obtaining the parameters of the Heisenberg Hamiltonian, we make larger supercells of intercalated PtSe₂ ($8 \times 8 \times 8$, or 512 magnetic sites), and study the magnetic phase transition using MC simulations. For each material, we perform ten independent MC runs, each with a different initial condition. We simulate the magnetic order using 3000 equilibration steps and 3000 subsequent MC steps. From the MC simulations, we obtain the specific heat and the magnetization of the intercalated PtSe₂ as a function of temperature. From the peak of the specific heat, we determine the critical (Curie/Néel) temperature.

3. Results and Discussion

3.1. Structure of Intercalated PtSe₂

Figure 2 shows the structure of intercalated PtSe₂. Figure 2a,b show the top and the side view of the most stable structure of the intercalated PtSe₂ for each of Ti, V, Cr, and Mn intercalation (see Section S2 of the Supplementary Information for structural details, and Section S3 for details on the formation energy calculations), obtained from DFT relaxation in a $2 \times 2 \times 1$ supercell.

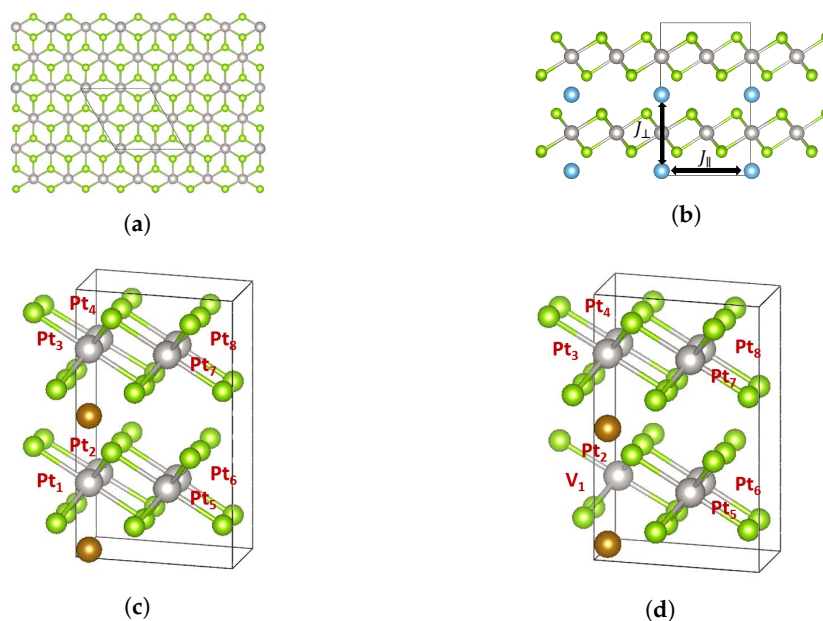


Figure 2. (a) Top view of the intercalated PtSe₂. The original unit cell has 8 Pt atoms, 16 Se atoms, and 2 intercalant atoms. (b) The side view of the intercalated PtSe₂. The in-plane (J_{\parallel}) and out-of-plane (J_{\perp}) exchange interactions are shown as well. (c) The PtSe₂ structure without vacancies. (d) The PtSe₂ structure with a Pt vacancy at the location of Pt₁.

We define the intercalation fraction according to the number of potential intercalant sites. The intercalation of 3d transition metals into MX₂ TMDs (with M = metal and X = chalcogen), such as PtSe₂ and WSe₂, typically happens at sites with octahedral coordination of X atoms [26]. Therefore, since we have one intercalant for every four octahedral intercalation sites, we express the intercalated PtSe₂ as TM_{1/4}PtSe₂ (where TM = intercalant atom). We use the same DFT-relaxed TM_{1/4}PtSe₂ in all subsequent calculations. We choose the fraction TM_{1/4} for our study because the intercalants are likely to form an ordered superlattice along the *c* axis of the hexagonal unit cells (i.e., in the out-of-plane direction) [26,27].

Figure 2c,d compare the PtSe₂ structure with and without the presence of the vacancy V₁ at the site of atom Pt₁. We find that the Pt vacancy causes the structure to change in the out-of-plane direction; see Table S1.

3.2. Magnetic Ordering and Critical Temperature

Figure 3a shows the magnetization as a function of temperature for intercalated PtSe₂ for intercalants Ti, V, Cr, and Mn without Pt vacancies. We normalize the curves to the saturation magnetization M_{sat} , which is the maximum magnetization that can be achieved in the material, when all magnetic moments point in exactly the same direction. We observe that when no vacancies are present, the magnetization for V saturates at low temperatures, suggesting a ferromagnetic transition. For Ti, Cr, and Mn, the magnetization vanishes at lower temperatures, suggesting an anti-ferromagnetic transition.

Figure 3b shows the magnetization as a function of temperature for intercalated PtSe₂ for intercalants Ti, V, Cr, and Mn with Pt vacancies. We observe that when the vacancies are present, the magnetization of Ti-, V-, and Mn-intercalated PtSe₂ goes to zero, suggesting an anti-ferromagnetic order. For Cr-intercalated PtSe₂, the magnetization reaches saturation, suggesting a ferromagnetic transition.

Figure 3c shows the specific heat as a function of temperature for intercalated pristine PtSe₂. We observe that the specific heat peaks at 111 K for V, 133 K for Cr, and at 68 K for Mn. However, for Ti, the specific heat peaks at a much lower temperature, namely below the lowest temperature point in our simulation, 0.01 K.

Figure 3d shows the specific heat as a function of temperature for intercalated pristine PtSe₂. When vacancies are present, the specific heat peaks at 63 K, 32 K, 221 K, and 45 K for Ti, V, Cr, and Mn, respectively.

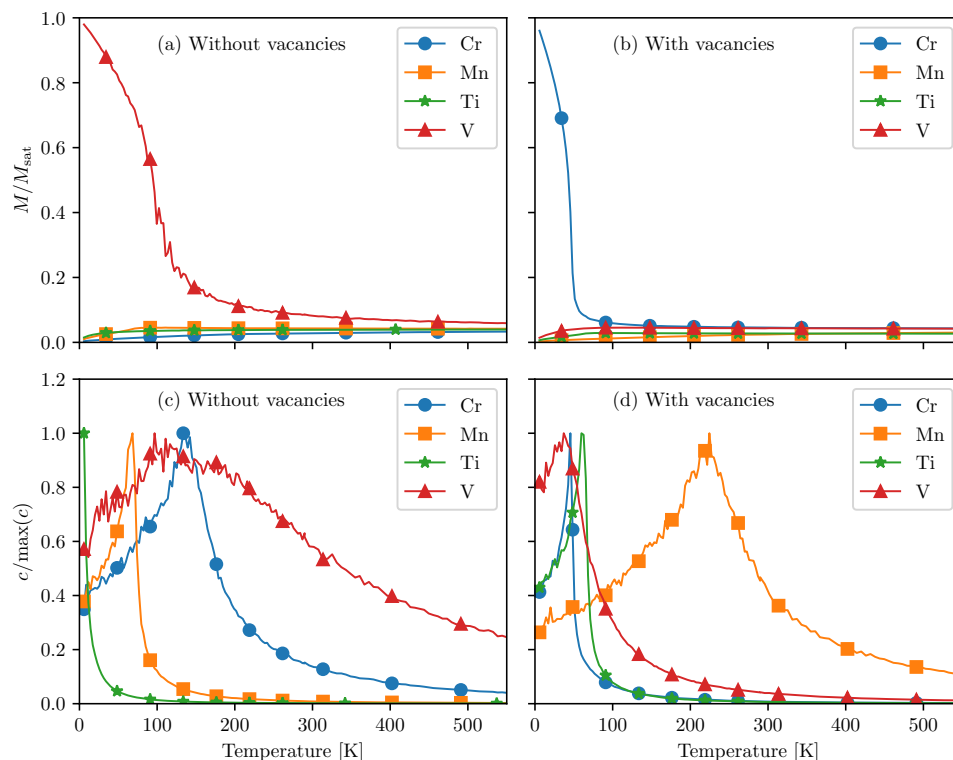


Figure 3. (a) The magnetization of intercalated PtSe₂ for various intercalants without any vacancies. M is the magnetization per intercalant atom obtained from the MC simulations and M_{sat} is the saturation magnetization. (b) The magnetization of intercalated PtSe₂ for various intercalants in the presence of vacancies. (c) The specific heat c of intercalated PtSe₂ for various intercalants without any vacancies. (d) The specific heat c of intercalated PtSe₂ in the case where vacancies are present.

3.3. Exchange Interactions and Magnetic Groundstate

Table 1 shows the obtained J parameters, onsite anisotropy (D), and the magnetic moment (M_{sat}) for various intercalants obtained from the DFT calculations, with and without Pt vacancies. The J parameters increase quite significantly for Ti and Mn, boosting their Néel temperature from below 0.01 K and 68 K in pristine PtSe₂ to 68 K and 221 K in defective PtSe₂, respectively. Moreover, we see that although the out-of-plane J parameters ($J_{\perp}^{x/z}$) of V remain positive, the in-plane J parameters ($J_{\parallel}^{x/z}$) become negative, resulting in an in-plane AFM and out-of-plane FM groundstate. The V-intercalated PtSe₂ therefore changes from a ferromagnet with T_{C} of 111 K, to an anti-ferromagnet with a T_{N} of 32 K. For Cr, all J parameters, both in-plane and out-of-plane, change sign, resulting in a change in the magnetic groundstate from purely anti-ferromagnetic to purely ferromagnetic, when vacancies are present. Cr-intercalated PtSe₂ changes from an anti-ferromagnet with $T_{\text{N}} = 133$ K when the PtSe₂ is pristine to a ferromagnet with $T_{\text{C}} = 45$ K when the material contains vacancies. Additionally, we find that for all the intercalants except for Ti, the out-of-plane interaction ($J_{\perp}^{x/z}$) is stronger than the in-plane interaction, suggesting a strong out-of-plane super super-exchange interaction [28].

The interaction between magnetic intercalant atoms occurs through a super super-exchange interaction chain X – Se – Pt – Se – X, where X denotes the intercalant atom (in our case: Ti, V, Cr, or Mn). The d orbitals of the intercalant atoms couple to each other through the p orbitals of the Se atoms and the d orbitals of the Pt atom in the chain. In the super super-exchange mechanism, each step in the chain consists of an anti-ferromagnetic

coupling, so that the X atoms are ferromagnetically coupled. In the case where the X atoms have a positive spin density, the chain consists of the following spin density signs: X (+) – Se (-) – Pt (+) – Se (-) – X (+), where “+” refers to a positive spin density, and “-” refers to a negative spin density. When such a ferromagnetic coupling is satisfied, the super super-exchange interactions will stabilize the ferromagnetic state and be the reason that the out-of-plane exchange parameters are large.

Table 1. Summary of the results for the intercalated PtSe₂ with and without vacancies. Results for structures with vacancies are in the columns marked with (Pt_v).

Intercalant Pt Vacancy Yes/No	Ti		V		Cr		Mn	
	No	Yes	No	Yes	No	Yes	No	Yes
Formation energy (eV)	-0.39	-2.09	0.77	-0.81	1.44	0.12	1.27	-0.06
<i>U</i> -value (eV)	4.23	3.86	4.45	4.01	5.03	3.72	6.80	5.18
Magnetic moment (μ_B)	1.52	0.60	2.77	2.6	3.90	3.55	4.44	4.26
J_{\perp}^z (meV/ μ_B^2)	0.05	-3.91	3.07	0.71	-0.65	0.04	-0.09	-0.82
J_{\parallel}^z (meV/ μ_B^2)	-0.26	2.79	0.07	-0.09	-0.20	0.25	0.07	-0.25
J_{\perp}^x (meV/ μ_B^2)	0.02	-5.16	3.25	0.72	-0.54	0.02	-0.10	-0.75
J_{\parallel}^x (meV/ μ_B^2)	-0.28	3.33	0.06	-0.10	-0.16	0.24	0.07	-0.20
<i>D</i> (meV/ μ_B^2)	-0.04	2.31	0.33	0.04	0.38	-0.02	-0.02	0.36
<i>T</i> _C (K)	—	—	111	—	—	45	—	—
<i>T</i> _N (K)	< 0.01	63	—	32	133	—	68	221

In the case of V-intercalated PtSe₂ without vacancies, we see that the V (+) – Se (-) – Pt (+) – Se (-) – V (+) chain is satisfied; see Figure S3a in the Supplementary Information. From Table 1, we see that the out-of-plane exchange coupling is indeed strong, $J_{\perp}^z = 3.07$ meV/ μ_B^2 and $J_{\perp}^x = 3.25$ meV/ μ_B^2 (see Table 1), when compared to the in-plane exchange interaction. Such a coupling through the super super-exchange mechanism stabilizes the ferromagnetic state, which is why the V-intercalated PtSe₂ without vacancies has a ferromagnetic groundstate.

For Mn-intercalated PtSe₂, however, we see that the chain has a different form: the Mn (+) – Se (-) – Pt (-) – Se (-) – Mn (+); see Figure S3b in the Supplementary Information. We see that the super super-exchange interaction does not take place, and that because of the lack of super super-exchange interaction, the out-of-plane exchange parameters are small, namely $J_{\perp}^z = -0.09$ meV/ μ_B^2 and $J_{\perp}^x = 0.07$ meV/ μ_B^2 ; see Table 1. The lack of stabilizing super super-exchange interactions causes an energy penalty in the ferromagnetic state, which pushes the state up in energy. The result is that the groundstate of Mn-intercalated PtSe₂ is not the ferromagnetic state, but an anti-ferromagnetic state.

Additionally, looking at the effect of vacancies, we can attribute the destabilization of the ferromagnetic state of V-intercalated PtSe₂ to a disruption of the super super-exchange interactions, causing a drop in the strength of the out-of-plane exchange parameters, namely J_{\perp}^z drops from 3.07 meV/ μ_B^2 to 0.71 meV/ μ_B^2 and J_{\perp}^x drops from 3.25 meV/ μ_B^2 to 0.72 meV/ μ_B^2 when the Pt vacancies are considered. The disruption of the super super-exchange mechanism causes the ferromagnetic state to shift up in total energy, and the groundstate goes from being a ferromagnetic state with a Curie temperature of 111 K to being an anti-ferromagnetic state with a Néel temperature of 32 K.

Comparing the *J* parameters in Table 1, we see that the sign of either the in-plane ($J_{\parallel}^{x/z}$) or the out-of-plane ($J_{\perp}^{x/z}$) exchange parameters change for all the intercalants. The change in the sign of *J* parameters suggests that the magnetic groundstate changes for all of the intercalated PtSe₂ with Pt vacancy compared to intercalated pristine PtSe₂. The tunability of *J* parameters with vacancy suggests that the magnetic properties of intercalated PtSe₂ can be tuned by creating vacancies in PtSe₂.

From the DFT results, we extract the localized magnetic moments on each atom of our intercalated materials. We find that the magnetic moments on the Pt and Se atoms are small compared to the moments on the intercalant atoms. The largest relative magnetic moment on a Pt atom occurs in Ti-intercalated PtSe₂ in the presence of vacancies and its size is 8.43% of that of the Ti atoms. The smallest relative magnetic moment on a Pt atom occurs in pristine Mn-intercalated PtSe₂ and its size is only 0.5% of the magnetic moment of the Mn atoms. Therefore, we conclude that our assumption that the magnetic moments are mostly located on the intercalant atoms is correct.

4. Conclusions

We have theoretically studied the possibility of realizing magnetic order in 1T-PtSe₂ through magnetic intercalation with Ti, V, Cr, and Mn. We showed that Ti results in an in-plane AFM and out-of-plane FM groundstate, whereas Mn results in in-plane FM and out-of-plane AFM. V results in an FM groundstate both in-plane and out-of-plane, whereas Cr results in an AFM groundstate both in the in-plane and out-of-plane direction. The critical temperatures that we find are lower than 0.01 K, 111 K, 133 K, and 68 K for Ti, V, Cr, and Mn, respectively.

We have further shown that the Pt vacancy significantly impacts the magnetic order in intercalated PtSe₂ both qualitatively and quantitatively. Most significantly, V intercalants become in-plane AFM from in-plane FM, and Cr intercalants transition from an AFM groundstate to an FM groundstate. Moreover, the Néel temperature of both Ti and Mn intercalants increases with Pt vacancy to 63 K and 221 K compared to a T_N lower than 0.01 K and $T_N = 68$ K in the pristine PtSe₂, respectively. Finally, we have shown that Pt vacancies can reduce the energy of formation in intercalated PtSe₂.

The tunability of the magnetic groundstate and critical temperature opens a plethora of opportunities for defect engineering the magnetic groundstate in PtSe₂ through intercalation. Further exploration of the electronic properties of intercalated PtSe₂ would provide deeper insights into the tuning of the magnetic order in PtSe₂. Additionally, we would like to mention that investigating the stability of magnetic states for different intercalant fractions would be an interesting avenue for future studies.

Supplementary Materials: The following are available online at <https://www.mdpi.com/article/10.3390/ma14154167/s1>, Figure S1: The supercell used in the next-nearest neighbor calculations, Table S1: The lattice vectors and angles for the intercalated PtSe₂ structures with and without Pt vacancies, Figure S2: The formation energy of the different intercalated PtSe₂ structures, with and without vacancies present, Figure S3: (a) Spin density for V-intercalated PtSe₂. (b) Spin density for Mn-intercalated PtSe₂.

Author Contributions: Conceptualization: W.G.V. and B.S.; Funding acquisition: W.G.V. and B.S.; Code development: S.T. and P.D.R.; Simulations: P.D.R.; Analysis: P.D.R., S.T., M.L.V.d.P.; Supervision: M.L.V.d.P., W.G.V., B.S.; Writing—Original draft: P.D.R.; Writing—Discussion and editing: All authors. All authors have read and agreed to the published version of the manuscript.

Funding: The project or effort depicted was or is sponsored by the Department of Defense, Defense Threat Reduction Agency. The content of the information does not necessarily reflect the position or the policy of the federal government, and no official endorsement should be inferred. This material is based upon work supported by the National Science Foundation under Grant No. 1802166. Any opinions, findings, and conclusions or recommendations expressed in this material are those of the author(s) and do not necessarily reflect the views of the National Science Foundation. This work was supported by imec's Industrial Affiliation Program.

Data Availability Statement: The data presented in this study are available within the article (and its supplementary material).

Acknowledgments: The authors acknowledge the Texas Advanced Computing Center (TACC) at The University of Texas at Austin for providing HPC resources that have contributed to the research results reported within this paper: <http://www.tacc.utexas.edu> (accessed on 21 July 2021). Peter

D. Reyntjens acknowledges support by the Eugene McDermott Fellowship program, under Grant Number 201806.

Conflicts of Interest: The authors declare no conflict of interest. The funders had no role in the design of the study; in the collection, analyses, or interpretation of data; in the writing of the manuscript, or in the decision to publish the results.

Abbreviations

The following abbreviations are used in this manuscript:

TMD	Transition Metal Dichalcogenide
DFT	Density Functional Theory
MC	Monte Carlo
FM	Ferromagnetic
AFM	Anti-ferromagnetic

Appendix A. Appendix: DFT Calculations

For all our DFT calculations, we use the Vienna Ab-initio Simulation Package (VASP) [29–32]. We use the Projector-Augmented Wave (PAW) method [33] and the generalized gradient approximation proposed by Perdew, Burke and Ernzerhof [34] for the exchange and correlation functionals in our calculations. To account for the Van der Waals interactions, we use the DFT-D3 method of Grimme et al. [35]. In all our DFT calculations, we set the plane-wave cutoff energy at 500 eV and employ an energetic convergence criterion of 10^{-5} eV to ensure our results are accurate. Prior to any self-consistent or Monte Carlo calculation, we perform the structural optimization of our systems by changing the lattice parameters and ionic positions until all the forces on the ions are lower than 0.005 eV/Å.

Appendix A.1. Hubbard U Correction

The intercalants are elements with highly correlated electrons in the 3d-shell. Because we use atoms which have an incomplete 3d-shell as intercalants, we need to consider the correlation effects of electrons in the 3d-shell. We use the Hubbard U correction within DFT (DFT + U) [36]. We estimate the value of the U parameter using the linear response method of M. Cococcioni and S. De Gironcoli [37]. For each PtSe₂ + intercalant system, we use the value of U obtained using the linear response method in all spin-polarized noncollinear DFT + U calculations for that system. The results are listed in Table 1.

Appendix A.2. Total Energies of the Magnetic States

The strength of magnetic interactions between the dopants depends on the relative stability of the various magnetic states. To get an accurate picture of the magnetic behavior around the critical (Curie or Néel) temperature, we therefore calculate the total energies of different ferromagnetic, ferrimagnetic and anti-ferromagnetic states for each material. On each intercalant site, we take all ferromagnetic, ferrimagnetic and anti-ferromagnetic combinations of the up and down spins states. Once the different spin configurations have been determined we perform non-collinear DFT + U by aligning each configuration along the [100] and [001] directions. We set the size of the initial magnetic moments in our DFT simulations according to the number of unpaired electrons in the d-shell of the intercalant atoms. The final groundstate magnetic moments are lower than the number of unpaired electrons due to the formation of bonds.

References

1. Cortie, D.L.; Causer, G.L.; Rule, K.C.; Fritzsche, H.; Kreuzpaintner, W.; Klose, F. Two-Dimensional Magnets: Forgotten History and Recent Progress towards Spintronic Applications. *Adv. Funct. Mater.* **2020**, *30*, 1901414. [[CrossRef](#)]
2. McGuire, M.A.; Dixit, H.; Cooper, V.R.; Sales, B.C. Coupling of Crystal Structure and Magnetism in the Layered, Ferromagnetic Insulator CrI₃. *Chem. Mater.* **2015**, *27*, 612–620. [[CrossRef](#)]

3. Huang, B.; Clark, G.; Navarro-Moratalla, E.; Klein, D.R.; Cheng, R.; Seyler, K.L.; Zhong, D.; Schmidgall, E.; McGuire, M.A.; Cobden, D.H.; et al. Layer-dependent ferromagnetism in a Van der Waals crystal down to the monolayer limit. *Nature* **2017**, *546*, 270–273. [[CrossRef](#)]
4. Gong, C.; Li, L.; Li, Z.; Ji, H.; Stern, A.; Xia, Y.; Cao, T.; Bao, W.; Wang, C.; Wang, Y.; et al. Discovery of intrinsic ferromagnetism in two-dimensional van der Waals crystals. *Nature* **2017**, *546*, 265–269. [[CrossRef](#)] [[PubMed](#)]
5. Torelli, D.; Thygesen, K.S.; Olsen, T. High throughput computational screening for 2D ferromagnetic materials: The critical role of anisotropy and local correlations. *2D Mater.* **2019**, *6*, 045018. [[CrossRef](#)]
6. Tiwari, S.; Vanherck, J.; de Put, M.L.V.; Vandenberghe, W.G.; Soree, B. Computing Curie Temperature of Two-Dimensional Ferromagnets in the Presence of Exchange Anisotropy. 2021. Available online: <https://arxiv.org/abs/2105.07958> (accessed on 21 July 2021).
7. Tiwari, S.; Van de Put, M.L.; Sorée, B.; Vandenberghe, W.G. Magnetic order and critical temperature of substitutionally doped transition metal dichalcogenide monolayers. *NPJ 2D Mater. Appl.* **2021**, *5*, 54. [[CrossRef](#)]
8. Reyntjens, P.D.; Tiwari, S.; de Put, M.L.V.; Sorée, B.; Vandenberghe, W.G. Magnetic properties and critical behavior of magnetically intercalated WSe₂: A theoretical study. *2D Mater.* **2020**, *8*, 025009. [[CrossRef](#)]
9. Habib, M.; Muhammad, Z.; Khan, R.; Wu, C.; ur Rehman, Z.; Zhou, Y.; Liu, H.; Song, L. Ferromagnetism in CVT grown tungsten diselenide single crystals with nickel doping. *Nanotechnology* **2018**, *29*, 115701. [[CrossRef](#)] [[PubMed](#)]
10. Yang, L.; Wu, H.; Zhang, W.; Lou, X.; Xie, Z.; Yu, X.; Liu, Y.; Chang, H. Ta Doping Enhanced Room-Temperature Ferromagnetism in 2D Semiconducting MoTe₂ Nanosheets. *Adv. Electron. Mater.* **2019**, *5*, 1900552. [[CrossRef](#)]
11. Mishra, R.; Zhou, W.; Pennycook, S.J.; Pantelides, S.T.; Idrobo, J.C. Long-range ferromagnetic ordering in manganese-doped two-dimensional dichalcogenides. *Phys. Rev. B* **2013**, *88*, 144409. [[CrossRef](#)]
12. Ramasubramaniam, A.; Naveh, D. Mn-doped monolayer MoS₂: An atomically thin dilute magnetic semiconductor. *Phys. Rev. B* **2013**, *87*, 195201. [[CrossRef](#)]
13. Yun, S.J.; Duong, D.L.; Ha, D.M.; Singh, K.; Phan, T.L.; Choi, W.; Kim, Y.M.; Lee, Y.H. Ferromagnetic Order at Room Temperature in Monolayer WSe₂ Semiconductor via Vanadium Dopant. *Adv. Sci.* **2020**, *7*, 1903076. [[CrossRef](#)]
14. Yue, Q.; Chang, S.; Qin, S.; Li, J. Functionalization of monolayer MoS₂ by substitutional doping: A first-principles study. *Phys. Lett. A* **2013**, *377*, 1362–1367. [[CrossRef](#)]
15. Luo, P.; Zhuge, F.; Zhang, Q.; Chen, Y.; Lv, L.; Huang, Y.; Li, H.; Zhai, T. Doping engineering and functionalization of two-dimensional metal chalcogenides. *Nanoscale Horiz.* **2019**, *4*, 26–51. [[CrossRef](#)] [[PubMed](#)]
16. Fei, Z.; Huang, B.; Malinowski, P.; Wang, W.; Song, T.; Sanchez, J.; Yao, W.; Xiao, D.; Zhu, X.; May, A.F.; et al. Two-dimensional itinerant ferromagnetism in atomically thin Fe₃GeTe₂. *Nat. Mater.* **2018**, *17*, 778–782. [[CrossRef](#)] [[PubMed](#)]
17. Guo, G.Y.; Liang, W.Y. The electronic structures of platinum dichalcogenides: PtS₂, PtSe₂ and PtTe₂. *J. Phys. C Solid State Phys.* **1986**, *19*, 995–1008. [[CrossRef](#)]
18. Wang, Y.; Li, L.; Yao, W.; Song, S.; Sun, J.T.; Pan, J.; Ren, X.; Li, C.; Okunishi, E.; Wang, Y.Q.; et al. Monolayer PtSe₂, a New Semiconducting Transition-Metal-Dichalcogenide, Epitaxially Grown by Direct Selenization of Pt. *Nano Lett.* **2015**, *15*, 4013–4018. [[CrossRef](#)] [[PubMed](#)]
19. Shawkat, M.S.; Gil, J.; Han, S.S.; Ko, T.J.; Wang, M.; Dev, D.; Kwon, J.; Lee, G.H.; Oh, K.H.; Chung, H.S.; et al. Thickness-Independent Semiconducting-to-Metallic Conversion in Wafer-Scale Two-Dimensional PtSe₂ Layers by Plasma-Driven Chalcogen Defect Engineering. *ACS Appl. Mater. Interfaces* **2020**, *12*, 14341–14351. [[CrossRef](#)] [[PubMed](#)]
20. Mermin, N.D.; Wagner, H. Absence of Ferromagnetism or Antiferromagnetism in One- or Two-Dimensional Isotropic Heisenberg Models. *Phys. Rev. Lett.* **1966**, *17*, 1133–1136. [[CrossRef](#)]
21. Kar, M.; Sarkar, R.; Pal, S.; Sarkar, P. Engineering the magnetic properties of PtSe₂ monolayer through transition metal doping. *J. Phys. Condens. Matter* **2019**, *31*, 145502. [[CrossRef](#)] [[PubMed](#)]
22. Avsar, A.; Cheon, C.Y.; Pizzochero, M.; Tripathi, M.; Ciarrocchi, A.; Yazyev, O.V.; Kis, A. Probing magnetism in atomically thin semiconducting PtSe₂. *Nat. Commun.* **2020**, *11*, 4806. [[CrossRef](#)] [[PubMed](#)]
23. Gao, J.; Cheng, Y.; Tian, T.; Hu, X.; Zeng, K.; Zhang, G.; Zhang, Y.W. Structure, Stability, and Kinetics of Vacancy Defects in Monolayer PtSe₂: A First-Principles Study. *ACS Omega* **2017**, *2*, 8640–8648. [[CrossRef](#)] [[PubMed](#)]
24. Zheng, H.; Choi, Y.; Baniasadi, F.; Hu, D.; Jiao, L.; Park, K.; Tao, C. Visualization of point defects in ultrathin layered 1T-PtSe₂. *2D Mater.* **2019**, *6*, 041005. [[CrossRef](#)]
25. Tiwari, S.; Van de Put, M.L.; Sorée, B.; Vandenberghe, W.G. Critical behavior of the ferromagnets CrI₃, CrBr₃, and CrGeTe₃ and the antiferromagnet FeCl₂: A detailed first-principles study. *Phys. Rev. B* **2021**, *103*, 014432. [[CrossRef](#)]
26. Friend, R.; Yoffe, A. Electronic properties of intercalation complexes of the transition metal dichalcogenides. *Adv. Phys.* **1987**, *36*, 1–94. [[CrossRef](#)]
27. Parkin, S.S.P.; Marseglia, E.A.; Brown, P.J. Magnetisation density distribution in Mn_{1/4}TaS₂: Observation of conduction electron spin polarisation. *J. Phys. C Solid State Phys.* **1983**, *16*, 2749–2764. [[CrossRef](#)]
28. Anderson, P.W. Antiferromagnetism. Theory of Superexchange Interaction. *Phys. Rev.* **1950**, *79*, 350–356. [[CrossRef](#)]
29. Kresse, G.; Furthmüller, J. Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set. *Comput. Mater. Sci.* **1996**, *6*, 15–50. [[CrossRef](#)]
30. Kresse, G.; Furthmüller, J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. *Phys. Rev. B* **1996**, *54*, 11169–11186. [[CrossRef](#)]

31. Kresse, G.; Hafner, J. Ab initio molecular dynamics for liquid metals. *Phys. Rev. B* **1993**, *47*, 558–561. [[CrossRef](#)]
32. Kresse, G.; Hafner, J. Ab initio molecular-dynamics simulation of the liquid-metal–amorphous-semiconductor transition in germanium. *Phys. Rev. B* **1994**, *49*, 14251–14269. [[CrossRef](#)] [[PubMed](#)]
33. Blöchl, P.E. Projector augmented-wave method. *Phys. Rev. B* **1994**, *50*, 17953–17979. [[CrossRef](#)]
34. Perdew, J.P.; Burke, K.; Ernzerhof, M. Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* **1996**, *77*, 3865–3868. [[CrossRef](#)] [[PubMed](#)]
35. Grimme, S.; Antony, J.; Ehrlich, S.; Krieg, H. A consistent and accurate ab initio parametrization of density functional dispersion correction (DFT-D) for the 94 elements H-Pu. *J. Chem. Phys.* **2010**, *132*, 154104. [[CrossRef](#)] [[PubMed](#)]
36. Dudarev, S.L.; Botton, G.A.; Savrasov, S.Y.; Humphreys, C.J.; Sutton, A.P. Electron-energy-loss spectra and the structural stability of nickel oxide: An LSDA+U study. *Phys. Rev. B* **1998**, *57*, 1505–1509. [[CrossRef](#)]
37. Cococcioni, M.; de Gironcoli, S. Linear response approach to the calculation of the effective interaction parameters in the LDA + U method. *Phys. Rev. B* **2005**, *71*, 035105. [[CrossRef](#)]