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CO Activation on TiO-Supported Cu and Ni Nanoclusters: Effect of Plasma-Induced Surface Charging

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CO₂ Activation on TiO₂-supported Cu₅ and Ni₅ Nanoclusters: Effect of Plasma-Induced Surface Charging

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Abstract

Surface charging is an often overlooked factor in many plasma-surface interactions and in particular in plasma catalysis. In this study, we investigate the effect of excess electrons induced by a plasma on the adsorption properties of CO₂ on titania-supported Cu₅ and Ni₅ clusters using spin polarized and dispersion corrected density functional theory (DFT) calculations. The effect of excess electrons on the adsorption of Ni and Cu pentamers as well as on CO₂ adsorption on a pristine anatase TiO₂ (101) slab is studied. Our results indicate that adding plasma-induced excess electrons to the system leads to a further stabilization of the bent CO₂ structure. Also dissociation of CO₂ on the charged clusters is energetically more favorable than on the neutral clusters. We hypothesize that surface charge is a plausible cause for the synergistic effects sometimes observed in plasma catalysis.

Introduction

Plasma catalysis^{1, 2} holds promise for many environmental³⁻⁸ and industrial^{9, 10} applications. Combining a plasma with a catalyst induces a complex reciprocal interaction, which in some cases leads to synergistic effects that improve the efficiency of the catalytic reactions. However, the underlying reasons for synergy in plasma catalysis remain unknown. Reaction promotion in catalytic removal of NO_x,¹¹ higher CO₂ conversion,¹² decrease in energy barrier for dry reforming of methane (DRM)¹³ and increased treatment efficiency on *Escherichia coli* cells in aqueous media¹⁴ have been reported as a result of synergistic effects in plasma catalysis.

Plasma is an ionized gas that consists of different interacting species such as electrons, ions, neutral atoms, and vibrationally and electronically excited species. The presence of these species makes the plasma a highly reactive but non-selective environment that is very suitable for combining with a catalyst.² In particular, the plasma leads to a modification of the gas phase (and hence the reactants), while still allowing for a (relatively) low temperature. The contribution of all the specific plasma factors strongly depends on the type of plasma and the working conditions, and has been studied extensively. The mechanisms of plasma-induced changes in the catalyst structural and electronic properties remain largely unresolved. In order to unravel (some of) the mechanisms underlying plasma-catalyst interactions we need to narrow down the complexity of the system by focusing on a specific factor and its effect on the whole plasma-catalytic process. Complementary to experiments, computer simulations are ideally suited for this.

A catalyst surface becomes inevitably charged upon interaction with plasma. The accumulated surface charge can alter the catalyst electronic and geometric structures, consequently leading to a change in the kinetics of the whole catalytic reaction. Even changes in activity of the different geometric structures of a catalyst are generally ascribed to differences in their electronic

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3 structures.¹⁵ Hence, by considering electronic properties of the catalyst during its interaction with
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5 plasma, one can gain information about the effect of plasma-induced surface charging on catalytic
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7 reactions.
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10 The effect of surface charging, for instance due to plasma, is important in heterogeneous
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12 catalysis and in particular when transition metal clusters are deposited on oxide surfaces. Excess
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14 electrons accumulated on the oxide surface can alter the electronic structure and adsorption
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16 properties of the supported transition metal nanoparticles. Although there have been
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18 experimental^{16, 17} and theoretical^{18, 19} investigations on the effect of plasma in heterogeneous
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20 catalysis, and studies have been carried out on the deposition of metal clusters on oxide surfaces
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22 without plasma,²⁰⁻²² the effect of plasma-induced excess electrons on the deposition of metal
23
24 clusters and their catalytic properties has been scarcely investigated. Both nickel and copper are
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26 widely used in heterogeneous catalysis, for instance, for CO₂ conversion^{18, 23, 24} e.g. for forming
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28 value-added products like methanol. We here choose to study Ni and Cu pentamers. First, these
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30 clusters can be synthesized and are actually used in the catalysis community. Further, because of
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32 their small size, they require less material, making them cheaper (per unit surface area) and
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34 possibly more environmentally friendly. Finally, considering the limited number of possible
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36 isomers, their fluxionality (in gas phase) is much less than larger clusters, which makes them
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38 computationally tractable.
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45 In this study we report on the role of excess electrons on (1) the adsorption of Ni₅ and Cu₅ clusters
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47 on an anatase TiO₂ (101) slab as test systems; (2) adsorption of an adsorbate molecule (CO₂) on
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49 the stoichiometric surface of anatase TiO₂ (101); (3) adsorption of a CO₂ molecule on TiO₂-
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51 supported Ni₅ and Cu₅ clusters; and (4) dissociation of the CO₂ molecule.
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3 As a reducible oxide, TiO_2 is an interesting candidate to investigate the effects of surface
4 charging. Among the various polymorphs of titania, anatase has been widely used in modeling
5 catalytic reactions.^{20-22, 25, 26} The (101) plane of anatase ($\alpha\text{-TiO}_2$) has been chosen for the adsorption
6 of the metal clusters because of its high thermodynamic stability.²¹
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15 **Computational details**

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17 All calculations have been carried out based on spin polarized density functional theory (DFT)
18 by using the Quickstep²⁷ module of the CP2K²⁸ package. This module mainly employs localized
19 Gaussian orbitals and auxiliary plane waves as a dual basis for expanding the Kohn-Sham orbitals.
20 Double- ζ valence plus polarization (DZVP) basis sets which are optimized from molecular
21 interactions (MOLOPT)²⁹ combined with a plane wave basis set with 800 Ry cutoff have been
22 used for the calculations. Goedecker-Teter-Hutter (GTH) pseudopotentials^{30, 31} are used for
23 describing the inner shell electrons; for C, O, Cu, Ti and Ni atoms, 4, 6, 11, 12 and 18 valence
24 electrons are explicitly considered, respectively. The general gradient approximation (GGA)
25 Perdew-Burke-Ernzerhof (PBE)³² functional is applied for treating electron exchange and
26 correlation effects. Dispersion interactions are included through Grimme's D3 approximation
27 along with Becke-Jonsson damping.^{33, 34} The Broyden-Fletcher-Goldfarb-Shanno (BFGS) scheme
28 is used for geometry optimization where a maximum force of 0.02 eV/Å has been set for the
29 relaxation of atoms. k-point sampling is limited to the Γ -point only and the Hirshfeld-I scheme³⁵
30 is used for charge analysis of interactions between the adsorbates and the slab surface.
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49 Lattice parameters of bulk anatase TiO_2 ($a = b = 3.7845 \text{ \AA}$ and $c = 9.5143 \text{ \AA}$)³⁶ were used
50 for creating the surface slabs. A slab of anatase (101) with dimensions of $10.239 \times 11.3535 \times 30.00$
51 \AA^3 (1×3 supercell) consisting of 6 (O-Ti-O) trilayers and containing 108 atoms (Fig. 1) was used
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3 for the calculations. To mimic the bulk-like structure we fixed the bottom layers of the support in
4 a first set of calculations. This, however, also leads to interference in the excess electron's
5 localization pattern. Thus, we also examined a fully relaxed structure, with no atoms frozen. The
6 results with the fixed layers are mainly provided in the supporting information (SI); we here mainly
7 focus on the results achieved without applying any constraint on relaxation of the atoms.
8 Calculations for the metal clusters and the CO₂ molecule in gas phase have been done in a box of
9 15 × 15 × 15 Å³ dimensions.

19 For the treatment of charged configurations, we applied our recently proposed method³⁷ suitable
20 for modeling plasma-surface interactions. In this approach, we use a positive countercharge
21 located far from the surface (in the z-direction) without assigning a local basis set to it. Thus, the
22 electron density is not allowed to be distributed around the ion but rather it will be localized on the
23 surface. This is an ideal approach for modeling the effect of plasma-induced excess electrons on
24 the surface of a catalyst, because it explicitly incorporates both the negative charging from excess
25 electrons, as well as polarization effects and electric fields arising from positively charged ions in
26 the plasma sheath region in the gas phase.

37 Electrostatic interactions were described with the Martyna-Tuckerman (MT) Poisson solver³⁸,
38 which requires the z-component of the cell size to be very large (100 Å) for charged configurations.
39 The MT solver allows to impose full periodicity along the surface slab (in the xy plane), mimicking
40 a semi-infinite catalyst surface, while simultaneously removing periodicity along the z direction.
41 As such, unphysical interactions of the slab with itself are avoided, which would otherwise result
42 in significant computational artefacts when dealing with charged surfaces³⁹.

51 We employed DFT+U^{40, 41} calculations for cross-checking the results, while accounting for the
52 strong on-site Coulomb interactions. By using this approach, the self-interaction effect for Ti d-

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3 orbitals will be corrected by Hubbard's U parameter. This parameter is set to 5 eV for Ti atoms.
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5 Considering the small number of atoms in clusters studied in this work, and negligible impact of
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7 their self-interaction on the results, no Hubbard parameter is applied to the d-orbitals of Ni and Cu
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9 atoms. Considering that DFT+U provides a better picture for electron localization on support
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11 cations, we have referred to the PDOS and spin density diagrams obtained by the DFT+U
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13 calculations.
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17 We have investigated the effect of charge in two ways: First, re-optimization of the neutral
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19 structure after adding an excess electron to the system, and second, optimization of the initially
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21 negatively charged configuration. After finding the most stable mode for both neutral and charged
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23 configurations, we have made a comparison between the corresponding adsorption energies and
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25 other properties for the neutral and negatively charged structures.
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29 Adsorption energies of clusters on anatase are calculated based on equation (1), where $E_{cluster}$
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31 is the total energy of the Ni₅/Cu₅ clusters, $E_{slab + cluster}$ is the total energy of the complex
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33 (consisting of cluster and surface) and E_{slab} is the total energy of the bare slab:
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$$E_{Adsorption} = E_{slab + cluster} - E_{cluster} - E_{slab} \quad (1)$$

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38 Similarly, the adsorption energy of CO₂ on the supported cluster is calculated as:
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$$E_{Adsorption} = E_{complex + molecule} - E_{complex} - E_{molecule} \quad (2)$$

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43 Thermal and zero-point energy calculations are not considered.
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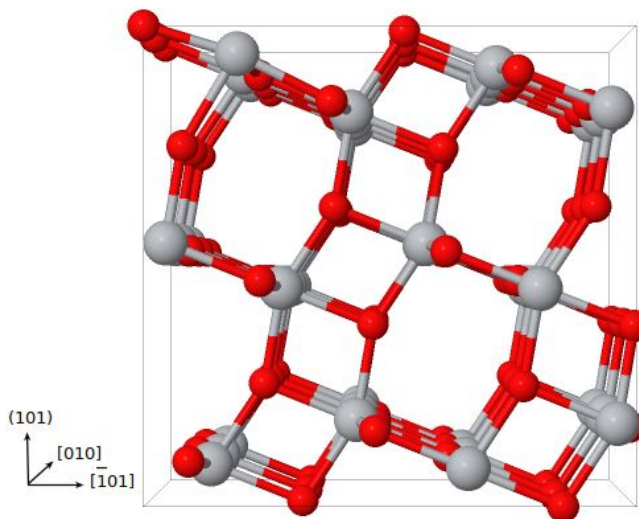


Figure 1: (1×3) supercell of anatase TiO_2 (101)

Results and discussion

Adsorption of Ni_5 and Cu_5 clusters on the anatase (101) surface

To investigate the adsorption of Cu_5 and Ni_5 on the TiO_2 surface, we first optimized their structure in the gas phase, as described in the SI. Then, the clusters were deposited on the neutral anatase (101) surface and their adsorption energy was calculated. Subsequently, one electron was added to this configuration, and the system was re-optimized. This allowed us to compare the adsorption properties of each metal cluster in the neutral and charged systems. Similar to the gas phase calculations, we explored 5 initial structures with different orientations of the cluster adsorbing on the surface. Then we examined the most stable orientation in 5 different adsorption sites. The most stable configurations found for the adsorption of Ni_5 and Cu_5 clusters are shown in Fig. 2 while the unstable adsorption modes are provided in the SI. We acknowledge that this procedure does not guarantee finding the global minimum.

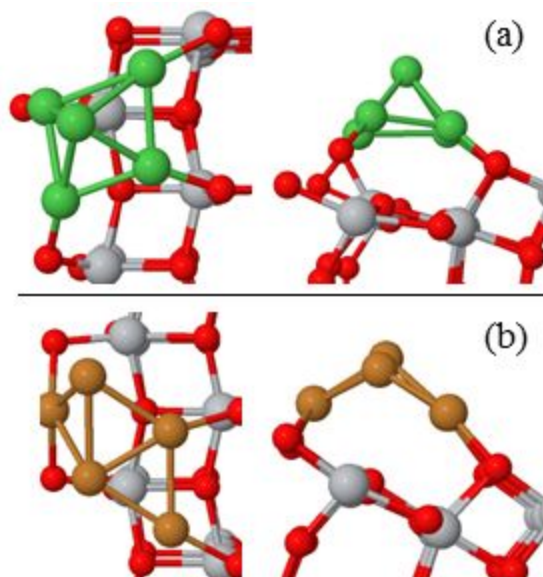


Figure 2: Top and side view of the most stable configurations of (a) Ni₅ and (b) Cu₅ clusters on anatase TiO₂ (101) slab

Upon adsorption, hardly any change in shape of the Ni₅ cluster is seen. Adsorption of the Ni₅ cluster on the anatase surface changes the electronic structure of the support such that the Ni d-orbitals fill the band-gap of the TiO₂ semiconductor (see the SI) in agreement with literature.^{42, 43} Population analysis based on the Hirshfeld scheme shows that during the adsorption of Ni₅ there is a charge transfer from the support to the cluster. For the gas phase Ni₅ cluster, the total spin moment ($n_{\alpha} - n_{\beta}$) is $2 \mu_B$ (triplet state) which changes to $3.1 \mu_B$ upon adsorption. Also the accumulated net charge of $-0.893 |e|$ on the Ni₅ cluster shows that there is a significant charge transfer between the cluster and the surface which leads to a high adsorption energy of -5.72 eV. This electron transfer from the support to the cluster leaves the support with one unpaired electron as can be seen from the spin density isosurfaces (Fig. 3a).

Upon adsorption, the Cu₅ cluster changes its original planar configuration to a more 3D structure facilitating strong interaction between the cluster and the oxide surface with an adsorption energy

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3 of -4.29 eV. In comparison to Ni_5 , the d-orbitals of the Cu_5 cluster partially lie in the TiO_2 band
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5 gap with a tendency to be located near the valence band (see the SI). Charge transfer direction
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7 between the support and cluster is different for Ni_5 and Cu_5 . The Cu_5 has a total spin moment of 2
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9 μ_B (doublet state) in the gas phase which changes to 0 after its adsorption on the support surface.
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11 During the adsorption, one unpaired electron in the Cu_5 cluster is transferred to the support. From
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13 Fig. 3(c) we can see that the transferred electron localizes on the subsurface Ti atom of the support.
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17 After adding an excess electron to the relaxed neutral structures and re-optimization, no change
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19 in the adsorption mode is seen. Changes in adsorption energies as described below can therefore
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21 (mostly) be attributed to the effect of the negative charge (Fig. 4). Adding a single excess electron
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23 to the surface model leads to an electron density of $8.6 \times 10^{17} \text{ m}^{-2}$ or a surface charge density of
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25 $-0.137 \frac{C}{\text{m}^2}$, which is in the same order of experimental estimates of plasma-deposited charge
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27 densities on dielectrics,⁴⁴ and a factor of four higher than reported in recent particle-in-cell/Monte
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29 Carlo collision PIC/MCC simulations for plasma in contact with dielectric surfaces or catalyst
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31 pores⁴⁵.
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37 From the spin density analysis (Fig. 3b) we can see that for the charged Ni_5 structure, the added
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39 electron is mostly localized on the Ni cluster, changing its net charge to $-1.643 |e|$ and also the
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41 total net spin moment to $2.2\mu_B$. This gives rise to a change in adsorption energy of the cluster by
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43 $+0.06$ eV, amounting to an adsorption energy of -5.66 eV. However, in Fig. 4, the DFT result
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45 without U correction shows that upon charging, the adsorption energy decreases by -0.51 eV,
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47 which is mostly because of the failure in localization of the excess electron.
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51 In contrast, upon negatively charging the surface, no spin polarization on the Cu_5 is seen, which
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53 means that the added electron is not localized on the cluster (i.e., the net spin moment of the cluster
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55 remains zero). Instead, as can be seen from Fig. 3d, the presence of excess electrons leads to a
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3 change in the localization of the previously transferred electron. Eventually, two electrons are
4 localized on the surface of the support, which decreases the adsorption energy by 1.20 eV for the
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6 Cu₅ cluster, amounting to an adsorption energy of -5.49 eV. A detailed summary of the changes
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8 in net spin moment and partial charges of the atoms for both neutral and charged structures is
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10 provided in the SI.
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15 These results highlight the different effect of excess charge on the adsorption properties of Cu
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17 and Ni pentamers. Methodologically, these results also show that depending on the localization
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19 procedure of the excess electron, its effect on stabilization of the adsorbed metal clusters can be
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21 different. Furthermore, it is seen that for fully relaxed systems, DFT calculations without
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23 considering on site Coulomb interactions have difficulties describing the effect of excess electrons
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25 on the stability of the metal clusters.
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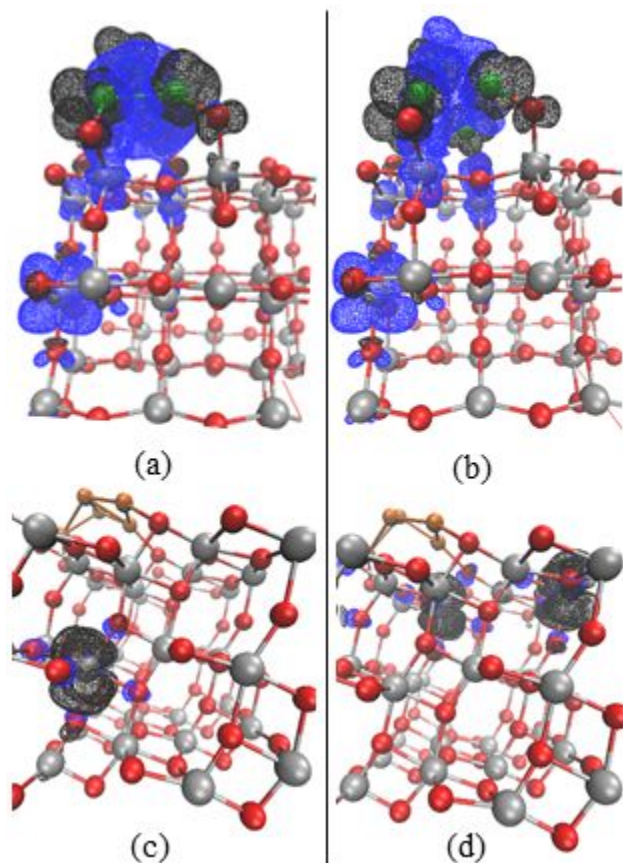


Figure 3: Spin density isosurfaces for a) neutral Ni₅/TiO₂, b) charged Ni₅/TiO₂, c) neutral Cu₅/TiO₂ and d) charged Cu₅/TiO₂ structures. The black and blue lobes correspond to spin up and spin down, respectively. Isovalues are set to $\pm 0.001 e/\text{\AA}^3$

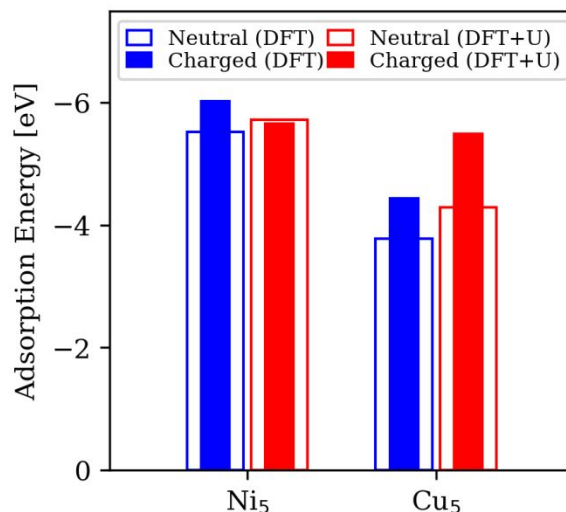


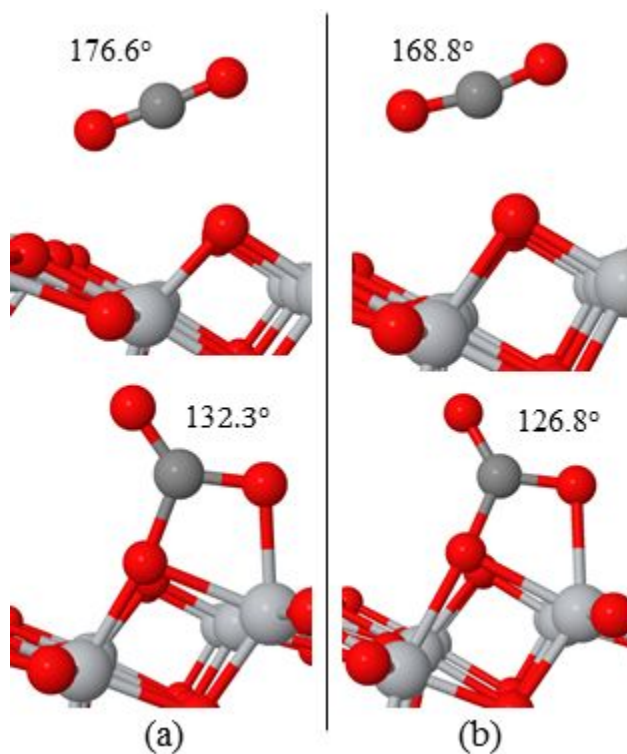
Figure 4: Adsorption energies of Ni₅ and Cu₅ clusters on anatase TiO₂ (101)

CO₂ adsorption on anatase TiO₂ (101)

CO₂ adsorption on neutral anatase was previously studied by DFT.^{43, 44, 46} In this study our aim is to investigate the effect of plasma-induced excess electrons on the adsorption of CO₂ on both the bare surface and on titania-supported Ni₅ and Cu₅ clusters. We will focus on the two most stable configurations for the adsorption of CO₂ on the anatase (101) surface, shown in Fig. 5.

We first kept the three layers of the TiO₂ slab frozen in their bulk structure to allow for a direct comparison between the results for neutral structures and the literature. For the linear configuration, we find that CO₂ adsorbs on the neutral surface with an adsorption energy of -0.44 eV and -0.51 eV for DFT and DFT+U, respectively. The bent structure shows an adsorption energy of -0.03 eV using DFT, while DFT+U yields an adsorption energy of -0.18 eV. It can be seen that the results coming from DFT+U are in agreement with the values of -0.48 eV for linear and -0.16 eV for carbonate-like structure, as reported by Sorescu et al.⁴⁴ Iyemperumal et al.⁴³ reported values of -0.40 eV and -0.09 eV for the most stable linear and bent structures,

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3 respectively. Results for both neutral and charged structures of CO₂ adsorption on bare TiO₂ (with
4 fixed bottom layers) are shown in Fig. 6a.



32 **Figure 5:** The most stable configurations for the linear and bent CO₂ structures adsorbing on
33 TiO₂ (101) surface. a) neutral and b) charged configurations

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37 By removing the constraint on the bottom layers of the slab, the electron distribution is unbiased,
38 which permits a reliable comparison between the neutral and charged structures. The results for
39 fully relaxed systems are provided in Fig. 6b. As we can see from this figure, the trends are
40 maintained, although there are quantitative differences in adsorption energies achieved for the
41 system with fixed layers.

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49 By adding charge to the system, both adsorption modes essentially maintain their structure. In
50 the linear adsorption mode, the effect of excess charge on the adsorption energy is less pronounced.
51 In this case, there is no mixing between the states of the molecule and the surface, such that the
52 (small) increase in adsorption energy of the linear CO₂ can be attributed to polarization effects. In

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3 this case electrostatic Coulomb forces between the positive C atom and the negatively charged
4 surface will lead to an improvement in adsorption energy. The PDOS of the linear adsorption mode
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6 surface will lead to an improvement in adsorption energy. The PDOS of the linear adsorption mode
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8 for both neutral and charged structures (Fig. 7a,7b) reveals that upon charging, the states of both
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10 oxygen atoms of the CO₂ molecule are shifted toward lower energies and this leads to a small
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12 improvement in stabilization of the molecule.
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15 For the bent structure there is a substantial increase in adsorption energy upon charging, to
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17 -1.13 eV and -1.40 eV in DFT and DFT+U, respectively, which is also reflected in the partial
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19 charges of CO₂ in the neutral and charged configurations: the partial charges on oxygen atoms of
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21 CO₂ molecule change from -0.862 |e| and -0.624 |e| (neutral system) to -1.028 |e| and
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23 -0.728 |e| (charged system). Considering that the surface Ti atom bonded to the oxygen atom of
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25 the CO₂ molecule has preserved its net charge from the neutral configuration, we can see that the
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27 increase in adsorption energy is mainly due to the change in partial charge of the molecule. This
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29 is consistent with the PDOS diagrams (Fig. 7d) of the charged configuration, showing a shift in
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31 the DOS of the oxygen atoms of CO₂ towards the valence band, facilitating the pronounced mixing
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33 with orbitals of the surface Ti atoms, in comparison to the neutral case.
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38 Table 1 shows the effect of charge on the adsorption energy of the two most stable modes of
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40 CO₂ on titania achieved by DFT and DFT+U. The same trend is clear from the results shown in
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42 Fig. 6.
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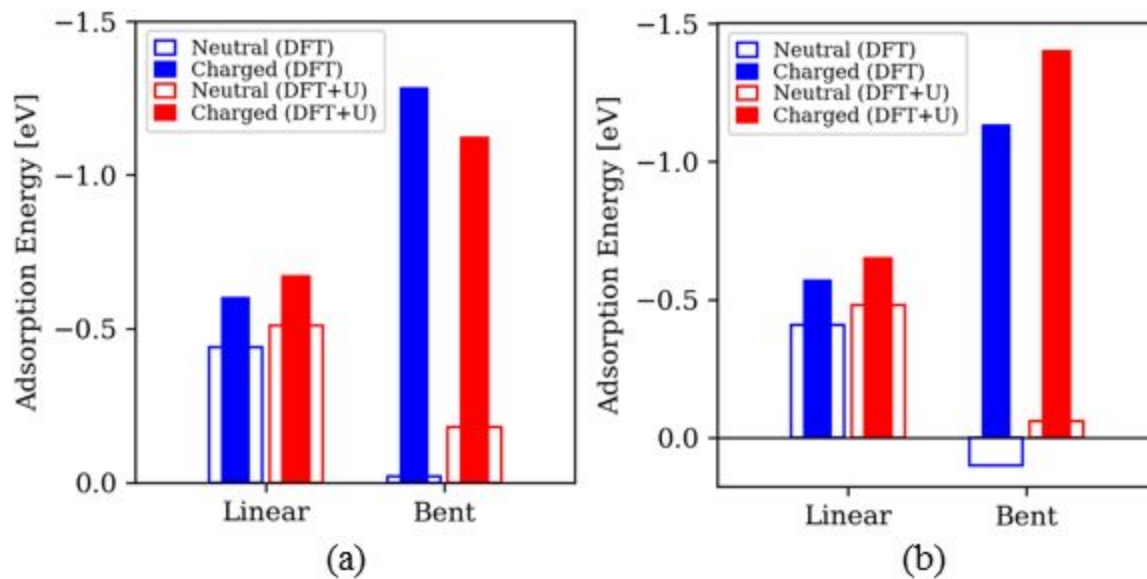


Figure 6: CO₂ adsorption energy for both linear and bent structures on neutral and charged configurations of anatase (101) - a) structures with 3 bottom layers kept frozen and b) fully relaxed structures

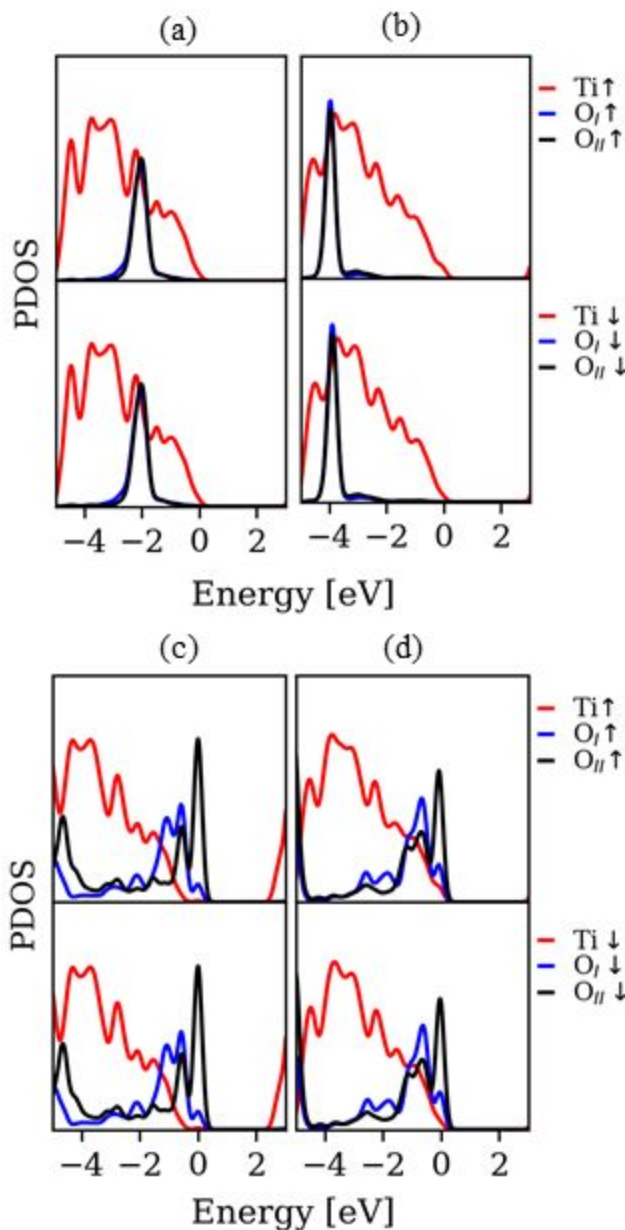


Figure 7: Projected density of states for the adsorption of a CO_2 molecule on TiO_2 . a) linear-neutral, b) linear-charged c) bent-neutral and d) bent-charged structures. O_I and O_{II} are oxygen atoms of the CO_2 molecule. Up/down spin states are shown with up/down arrows.

Transformation of the linear to the bent structure thus seems essential for CO_2 activation. It is clear from the results that adding a plasma-deposited excess electron to the system changes the most stable structure of the CO_2 adsorbed on titania surface from a linear to a bent configuration,

which makes the activation of the CO₂ molecule thermodynamically more favorable. Correspondingly, we also analyzed changes in C-O bond length and in O-C-O bond angle, as a result of negatively charging the surface. In Table 1, we summarize the results for the effect of charge on bond lengths and angles. It can be seen that the presence of excess charge has impact on bending the CO₂ molecule, while the effect on C-O bond elongation is limited.

Table 1: Bond lengths and adsorption energies of CO₂ with both linear and bent structures in neutral and charged configurations

Method	Structure	r(C-O)	α (OCO)	E _{ads} (eV)
	Gas Phase	1.17, 1.17	180	
DFT	Bent (Neutral)	1.20, 1.32	131.6	0.10
	Bent (Charged)	1.23, 1.32	126.4	-1.13
	Linear (Neutral)	1.17, 1.18	177.2	-0.41
	Linear (Charged)	1.18, 1.18	169.9	-0.57
	Bent (Neutral)	1.20, 1.32	132.3	-0.06
DFT+U	Bent (Charged)	1.23, 1.32	126.8	-1.40
	Linear (Neutral)	1.17, 1.18	176.6	-0.48
	Linear (Charged)	1.18, 1.18	168.8	-0.65

CO₂ adsorption on supported clusters

For the adsorption of CO₂ on titania-supported clusters, various initial modes were examined. In this case, we optimized each initial structure both with and without the presence of plasma-induced excess electrons.

Adsorption of CO₂ on the supported Ni₅ cluster does not lead to a marked change in shape of the cluster. Bond formation between the CO₂ molecule and Ni atoms of the cluster results in a bent structure of CO₂ (Fig. 8a) with an adsorption energy of -0.53 and -0.31 eV for DFT and DFT+U, respectively. During CO₂ adsorption, the net spin moment of the cluster remains at $3 \mu_B$, while its net charge increases from $-0.893 |e|$ to $-1.412 |e|$, indicating the formation of a polar covalent bond between the cluster and the C atom of the CO₂ molecule. The Ni atom which bonds to C has a net charge of $-0.730 |e|$, while the carbon atom has a charge of $1.769 |e|$.

Adsorption of CO₂ on the supported Cu₅ cluster results in a noticeable change in the cluster shape. In this case, CO₂ adsorbs on the interface via the C atom making bonds with two copper atoms of the cluster and the oxygen atom with a surface Ti atom (Fig. 8c). In this configuration, the Cu₅ cluster has three coordinations with surface oxygens and three with the CO₂ molecule, while maintaining a net spin moment of 0 and having partial net charge of $-0.205 |e|$. The net charge for the carbon and the two Cu atoms of the cluster is $1.502 |e|$, $-0.064 |e|$ and $-0.228 |e|$, respectively. This shows the formation of a covalent bond between the cluster and the CO₂ molecule.

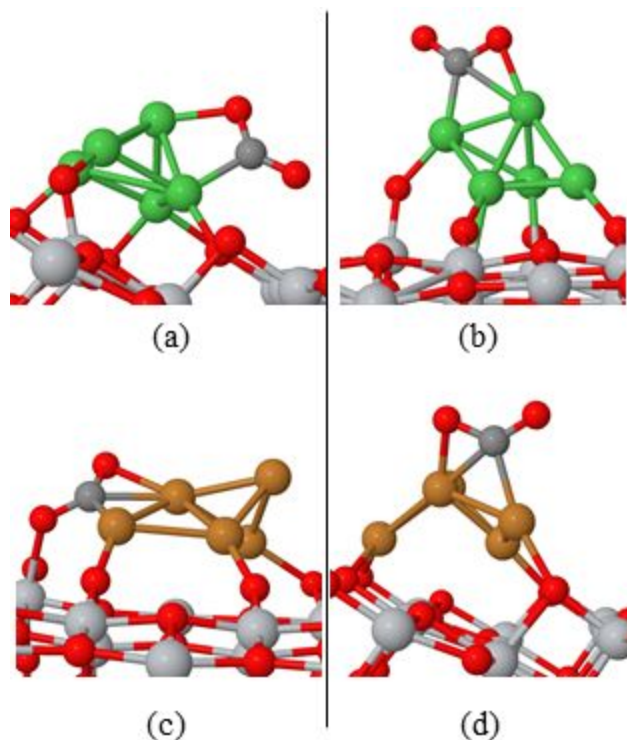


Figure 8: Adsorption of CO_2 on supported a) Ni_5 - neutral, b) Ni_5 - charged, c) Cu_5 - neutral and d) Cu_5 - charged structures

Bond lengths and adsorption energies of CO_2 on Ni_5 and Cu_5 clusters supported by anatase TiO_2 (101) are shown in Table 2. From this table and also from Fig. 9, we observe a noticeable improvement in stability of the CO_2 molecule adsorbed on both clusters upon negatively charging the surface.

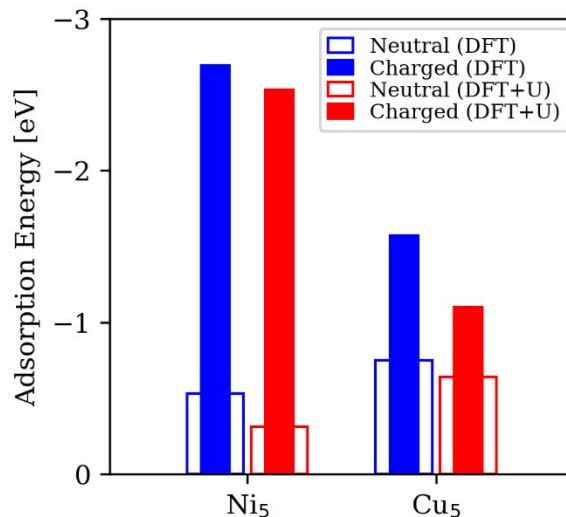
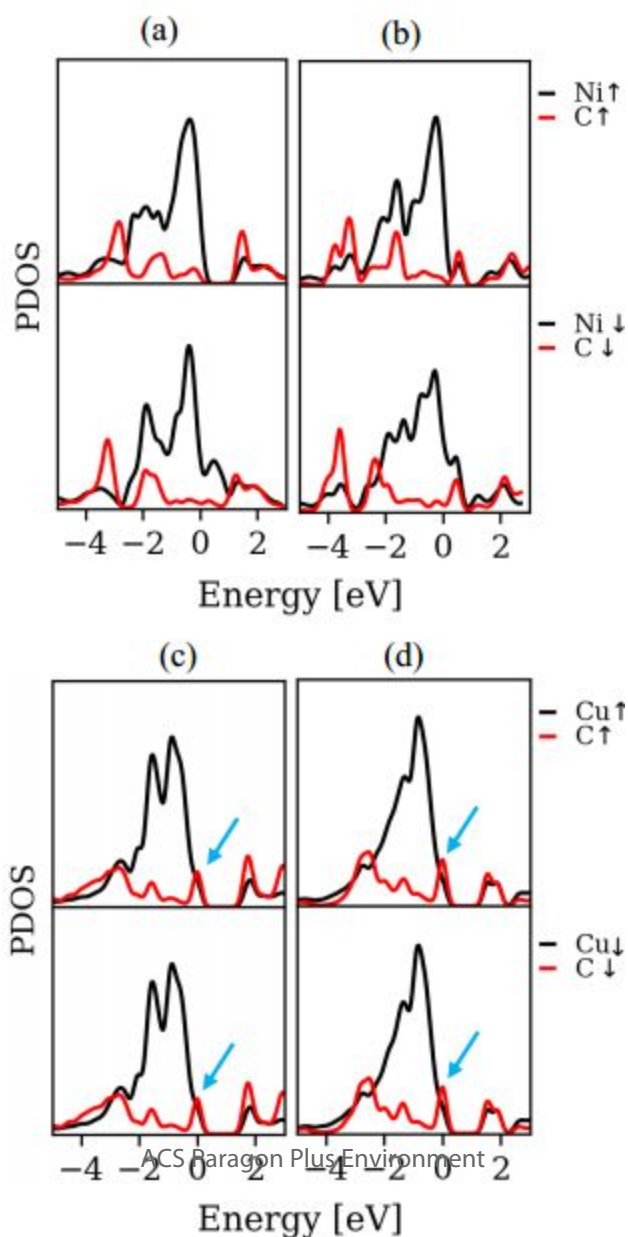


Figure 9: Adsorption energy of CO₂ molecule on Ni₅/Cu₅ clusters supported by anatase TiO₂ (101)

The increase in adsorption energy in case of the charged Ni₅ structure is mostly because of the extra bond formation between the C atom and the top Ni atom of the cluster (Fig. 8b). Together with a net spin moment of $2 \mu_B$, the net charge on the cluster increases to $-1.708 |e|$, explaining the stronger bond between the cluster and the (positively charged) C atom of the molecule. The PDOS reveals that upon charging there is a shift in bonding states of the CO₂ molecule towards the valence band (Fig. 10b). This leads to facilitation of the hybridization of high-lying CO₂ antibonding π^* orbitals with d orbitals of the cluster. This shift is also seen in CO₂ adsorption on a negatively charged slab of titania, which resulted in improved adsorption energy of the carbonate-like structure.

For the CO₂ adsorption on the negatively charged Cu₅/TiO₂ (Fig. 8d), we can see a slightly smaller bandgap (Fig. 10d), together with higher hybridization of C states with the cluster d orbitals. By analyzing the spin population and net charge of each atom, together with the morphological changes in the cluster shape upon charging the system, we can see that for the charged configuration, the CO₂ molecule binds only with the metal cluster, in contrast to the

neutral case, in which CO₂ binds to the interface. This change in adsorption mode is accompanied by a change in the amount of accumulated charge on the CO₂ molecule (from +0.184 |e| to -0.268 |e| upon negatively charging the system). The net spin moment remains zero, while the net charge on the cluster increases as a result of the presence of the excess electron. Considering the unpolarized spin configuration of the cluster, we conclude that the net charge of the cluster is actually the amount of charge that it shares with the surface as a result of their interaction. Thus, the improved adsorption energy of the CO₂ molecule on negatively charged Cu₅/TiO₂ is actually due to more hybridization of the CO₂ states with cluster's orbitals and also stronger interaction of the complex of cluster + molecule with support's surface (considering that the amount of charge shared between the cluster and the support has been increased).



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35 **Figure 10:** Projected density of states for the adsorption of CO₂ molecule on TiO₂ supported: a)
36 Ni₅-neutral, b) Ni₅-charged, c) Cu₅-neutral, d) Cu₅-charged structures
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46 **Table 2:** Bond lengths and adsorption energies of CO₂ molecules on TiO₂-supported Ni₅ and Cu₅
47 clusters
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Method	Structure	r(C-O)	α (OCO)	E _{ads} (eV)
DFT	Gas Phase	1.17, 1.17	180	

	Ni ₅ -Neutral	1.22, 1.27	135.9	-0.53
	Ni ₅ -Charged	1.26, 1.32	124.2	-2.69
	Cu ₅ -Neutral	1.25, 1.28	131.0	-0.75
	Cu ₅ -Charged	1.25, 1.30	125.6	-1.57
	<hr/>			
DFT+U	Ni ₅ -Neutral	1.23, 1.27	134.3	-0.31
	Ni ₅ -Charged	1.26, 1.32	123.7	-2.53
	Cu ₅ -Neutral	1.25, 1.29	129.5	-0.64
	Cu ₅ -Charged	1.25, 1.29	125.8	-1.10

By looking into the bond lengths and angles reported in Table 2, we can see that the bond length elongation and bending of the O-C-O angle in the Ni₅ case is more sensitive to excess charge than in the Cu₅ case. This agrees with the more stable bent structure of the CO₂ adsorbed on supported Ni₅.

Dissociated CO₂ on TiO₂, Ni₅/TiO₂ and Cu₅/TiO₂

In plasma catalysis CO₂ dissociation can occur via several reaction paths. In this work, we focus on the adsorption of CO₂ on neutral and negatively charged surfaces, and not on the actual reaction paths. Hence, these simulations do not aim to identify the most suitable catalyst for CO₂ dissociation, but they provide more insight in the effect of surface charging, which is one of the characteristic features of plasma catalysis, distinguishing it from thermal catalysis. In order to estimate the effect of charge on the dissociation of CO₂, we examined various initial configurations of dissociated CO₂ (CO(ads) + O(ads)) on TiO₂, Ni₅/TiO₂ and Cu₅/TiO₂.

For CO₂ dissociation on neutral TiO₂, we did not find a stable structure for adsorbed CO(ads). In most cases, CO either desorbs back in the gas phase or attracts an oxygen atom from the surface and so forms a linear CO₂ molecule. We have provided a less stable structure in Fig. 11a, in which

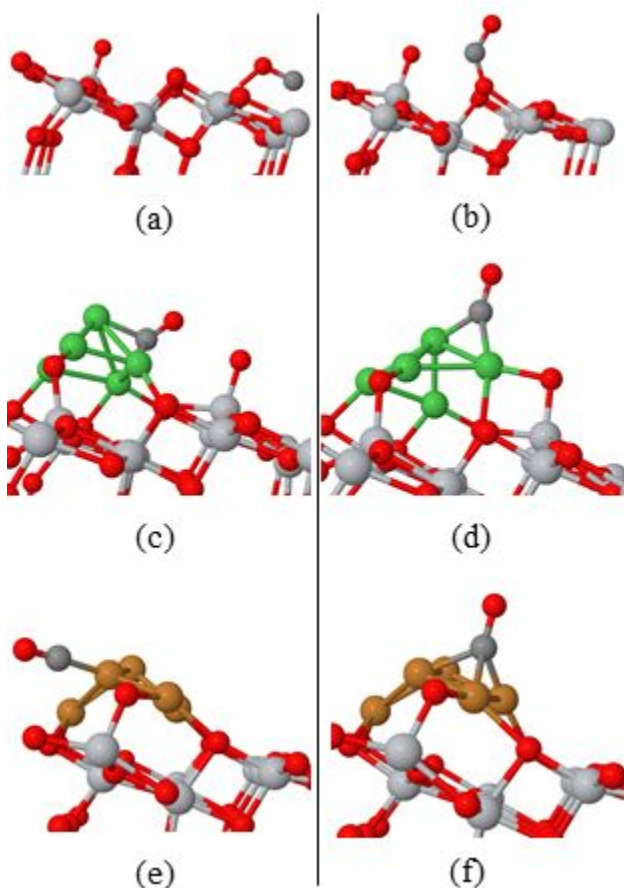
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3 the CO(ads) + O(ads) phase is achieved. Although results in Fig. 12 for dissociation on the bare
4 surface of TiO₂ show an improvement in the adsorption energy due to the presence of the excess
5 electron, dissociative adsorption is thermodynamically unfavorable, regardless of charging. This
6 means that in the absence of a catalyst, dissociation of CO₂ on a bare slab of anatase will not
7 proceed. An interesting parallel can be drawn with the effect of vacancies on the reactivity of TiO₂
8 towards CO₂ activation. Just like the formation of vacancies, electron addition effectively
9 corresponds to a reduction of the TiO₂ surface. In both cases, this reduction has been found to
10 result in an improved chemisorption of CO₂. However, vacancies additionally provide a strong
11 binding site for oxygen^{44, 47}, thus also promoting the dissociation of CO₂ adsorbed on a vacancy;
12 such site is not created upon charging, and CO₂ splitting remains unfavorable.
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26 By employing Ni₅ as catalyst, we find a dissociative adsorption energy of -0.03 eV for the CO₂
27 dissociation on the neutral surface (Fig. 11c). By adding an excess electron to the system the
28 adsorption energy significantly increases to -1.85 eV, which makes the reaction
29 thermodynamically highly favorable (Fig. 11d).
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35 For the case of neutral Cu₅/TiO₂, dissociation of the CO₂ molecule into adsorbed (CO)_{ads} and
36 (O)_{ads} is slightly endothermic with an adsorption energy of $+0.14$ eV (Fig. 11e). Upon negatively
37 charging the system, (Fig. 11f) the dissociative adsorption energy becomes -0.25 eV, making
38 this reaction thermodynamically favourable. Fig. 12 shows the corresponding dissociative
39 adsorption energy for each configuration based on DFT and DFT+U calculations.
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47 These results show that the presence of an excess electron has a positive impact on stabilization
48 of the dissociated CO₂. This impact is more pronounced in case of a Ni pentamer as catalyst, which
49 shows excellent performance in response to the presence of a plasma-induced excess electron and
50 significantly improves the adsorption energy of both adsorbed and dissociated CO₂. Cu₅ adsorbed
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3 on a neutral TiO_2 surface is not active enough to dissociate the CO_2 in a favorable way, but it also
4 shows positive feedback to negatively charging the surface. Therefore, these two particular test
5 cases (adsorbed Ni_5 and Cu_5) are good examples of how electrons from a non-equilibrium plasma
6 can induce a shift in the chemical properties of different catalysts.
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41 **Figure 11:** The most stable dissociated structures for CO_2 ($\text{CO(ads)} + \text{O(ads)}$) on a) neutral TiO_2 ,
42 b) charged TiO_2 , c) neutral Ni_5/TiO_2 , d) charged Ni_5/TiO_2 , e) neutral Cu_5/TiO_2 and f) charged
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46 Cu_5/TiO_2
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49 While there are small differences in bond lengths and also adsorption energies in the results
50 obtained from DFT and DFT+U, trends regarding the effect of charge on CO_2 activation are the
51 same for both approaches. In addition to the comparison between the DFT and DFT+U results, we
52 have also investigated the effect of change in surface charge density. We did the same calculations
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for CO₂ adsorption on pristine titania by using a larger slab (2 × 4 supercell) of anatase TiO₂ (101) corresponding to a surface charge density of $-0.05 \frac{C}{m^2}$ which is 2.7 times less than the small slab.

The results indicate that for the large slab, the trends in the effect of charge on stabilization of the bent CO₂ molecule remain the same, although less pronounced (see the SI).

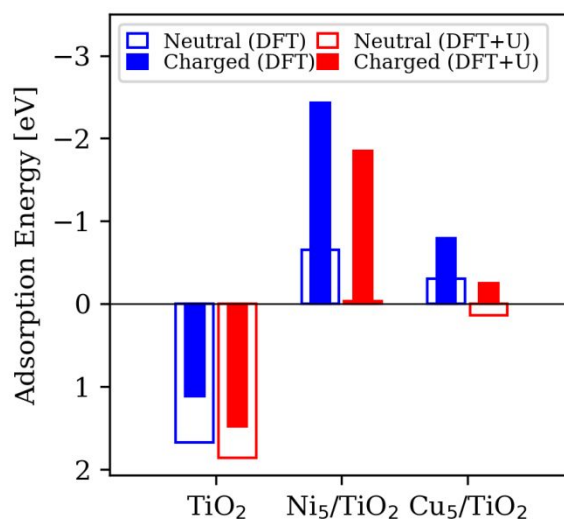


Figure 12: Dissociative adsorption energy of CO₂ on bare TiO₂, Ni₅/TiO₂ and Cu₅/TiO₂

Conclusion

In this study, we investigated the effect of excess electrons induced by plasma on the adsorption properties of Ni₅ and Cu₅ clusters over the anatase TiO₂ (101) surface and also CO₂ activation on both pristine a-TiO₂ and supported clusters.

Adsorption of Ni₅ and Cu₅ clusters on the surface of anatase (101) in the neutral case leads to a charge transfer between the cluster and the support. Depending on the type of catalyst, the direction of charge transfer can be from surface to the cluster and vice versa. In this case, negatively charging the system leads to minor changes in stabilization of the clusters. If the excess electron is localized on the surface, it will have a more pronounced impact than when the electron is localized on the

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3 cluster. Negatively charging the surface for adsorption of the CO₂ on a bare slab of anatase (101)
4 causes the most stable structure of the CO₂ molecule to change from a linear to a bent
5 configuration. This is mostly because of shifting the high-lying CO₂ antibonding π^* orbitals
6 towards the valence band and facilitating their hybridization with *p* or *d* states of the surface.
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12 In the case of employing Cu₅ and Ni₅ nanoclusters, the same result is seen for the improvement
13 of bent CO₂ stabilization, although the underlying mechanisms are slightly different. Upon
14 negatively charging the system in CO₂ adsorption on supported Ni₅, the same shifting in CO₂ high-
15 lying states is seen, which leads to increased hybridization with the cluster d orbitals. In contrast,
16 for the supported Cu₅, changing the adsorption site from the interface to the cluster leads to more
17 stabilization of the CO₂ molecule. This change in adsorption mode results in a significant increase
18 in the negative partial charge of the CO₂ molecule, which can be considered as indicative for
19 activation of the molecule⁴³.
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31 The dissociation of CO₂ on a bare slab of anatase titania is thermodynamically unfavourable,
32 regardless of added charge. CO₂ dissociation is favourable on Cu₅ as catalyst in a charged
33 configuration, and on Ni₅ both in neutral and charged configurations. Together with better results
34 for adsorption energy and bond elongation of the CO₂ (ads) in charged configurations, Ni₅ could
35 be a good option for plasma catalysis aiming to CO₂ conversion.
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42 In general, plasma-induced excess electrons alter the adsorption process, by shifting the anti-
43 bonding states of CO₂ towards the valence band, increasing the polarization effects and changing
44 the adsorption site of the molecule. These effects consequently lead to improved stabilization of
45 bent (ads) and dissociated CO₂ in negatively charged structures. This can be considered as a
46 possible reason for the synergistic effects in plasma catalysis.
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53 **Associated Content**

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3 The supporting information is available free of charge.
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5 Isolated Ni₅ and Cu₅ clusters, unstable structures for the Ni₅ and Cu₅ cluster adsorption on TiO₂
6 surface, PDOS plots for adsorption of clusters on TiO₂ (101), results for structures with fixed
7
8 bottom layers, calculations with (2×4) supercell of TiO₂ (101) (constrained and fully relaxed
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10 structures) and tables of spin moments and net charges.
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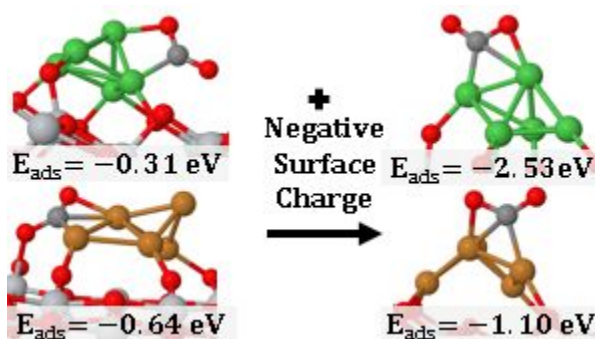
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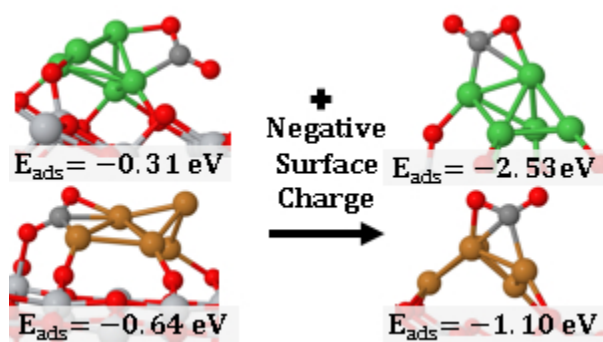
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TOC Graphic

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TOC

79x44mm (96 x 96 DPI)